## **Observation of All the Intermediate Steps of a Chemical Reaction on an Oxide Surface by Scanning Tunneling** Microscopy

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ne of the "Holy Grails" within the area of surface science is to directly "watch" chemical reactions at the atomic scale, and STM is the technique of choice for reaching this goal. In the past, the STM has been successfully used to manipulate matter on the atomic scale with single bond precision<sup>1-4</sup> to gain insight into self-diffusion,<sup>5,6</sup> the diffusion of small molecules and adatoms,<sup>7-11</sup> as well as into the diffusion of large organic molecules.<sup>12,13</sup> Moreover, the STM has successfully been used to study spontaneous dissociation reactions of small molecules<sup>14–16</sup> and reactions induced by the tunneling electrons from the STM tip.<sup>4,8,9</sup> For example, the dissociation of O<sub>2</sub> on Pt(111) has been studied by precisely positioning of the STM tip above the O<sub>2</sub> molecule and applying a voltage pulse.<sup>4</sup> However, STM studies unraveling complete surface-catalyzed reactions with all the intermediates are very scarce.

ABSTRACT By means of high-resolution scanning tunneling microscopy (STM), we have revealed unprecedented details about the intermediate steps for a surface-catalyzed reaction. Specifically, we studied the oxidation of H adatoms by 02 molecules on the rutile TiO2(110) surface. 02 adsorbs and successively reacts with the H adatoms, resulting in the formation of water species. Using time-lapsed STM imaging, we have unraveled the individual reaction intermediates of HO<sub>2</sub>, H<sub>2</sub>O<sub>2</sub>, and H<sub>3</sub>O<sub>2</sub> stoichiometry and the final reaction product—pairs of water molecules, [H<sub>2</sub>O]<sub>2</sub>. Because of their different appearance and mobility, these four species are discernible in the time-lapsed STM images. The interpretation of the STM results is corroborated by density functional theory calculations. The presented experimental and theoretical results are discussed with respect to previous reports where other reaction mechanisms have been put forward.

KEYWORDS: titania (TiO<sub>2</sub>) · oxygen · water · HO<sub>2</sub>/H<sub>2</sub>O<sub>2</sub> intermediates, hydroxyls · scanning tunneling microscopy (STM) · density functional theory (DFT)

Remarkable examples demonstrating how STM can be used to visualize the reactions on metal surfaces have been published by the group of Ertl and Wintterlin.<sup>11,17,18</sup> For instance, Sachs et al. studied the catalytic oxidation of hydrogen on Pt(111) and revealed the atomic processes in propagating reaction fronts.<sup>17</sup> For reactions on oxide sur-

faces, only few such studies

exist.<sup>19-22</sup> With a focus primarily on single crystals, we can

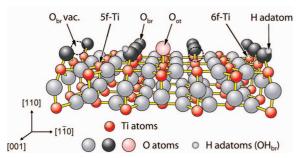


Figure 1. Ball-and-stick model of the TiO<sub>2</sub>(110)-(1  $\times$  1) surface with some of its known point defects. Large gray balls represent O atoms, and medium size red balls six-fold coordinated Ti (6f-Ti) and five-fold coordinated surface Ti atoms (5f-Ti), respectively. Small light gray balls represent H adatoms. The bridge-bonded O species (O<sub>br</sub>), single oxygen vacancies (O<sub>br</sub> vac.), and the on-top bonded O species (O<sub>ot</sub>) are also indicated.

restrict ourselves to the very intensely investigated (110) surface of rutile TiO<sub>2</sub> (Figure 1), whose investigations have been triggered through its ease in preparation and nufbe@inano.dk. merous applications.<sup>21,23–25</sup> Time-lapsed STM images-in

the following referred to as "STM movies"—acquired on this surface at high temperatures provided important information about self-diffusion

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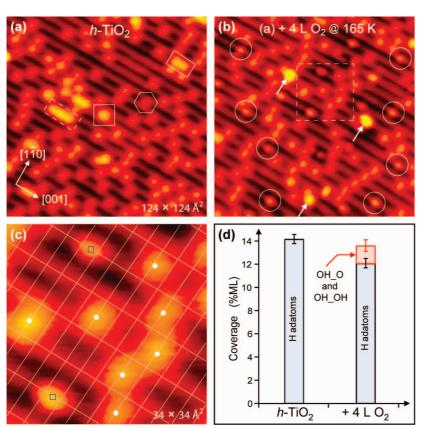


Figure 2. STM images of the *h*-TiO<sub>2</sub>(110) surface before (a) and after 4 L O<sub>2</sub> exposure at  $\sim$ 165 K (b). Symbols in (a) indicate single H adatoms (hexagon), paired H adatoms (square), next-nearest H adatoms (rectangle, full line), and a chain of next-nearest H adatoms (rectangle, broken line). The circles in (b) indicate the newly formed species in the Ti troughs, and arrows indicate larger protrusions in the Ti troughs of STM heights as typically observed for water species. The indicated square area in (b) (white broken line) is shown in (c) enlarged. A lattice grid was centered on top of 5f-Ti sites in (c) using single H adatoms (white small dots) for aligning in the [001] direction. Note that the z contrast in the STM image depicted in (c) is higher than that in the STM images shown in (a) and (b). The STM images were acquired at 110 K. (d) Bar graph of measured coverages in %ML of the obtained surface species before and after O<sub>2</sub> exposure. To extract the densities of surface species, areas of ~2000 nm<sup>2</sup> were scanned and analyzed.

or the reoxidation of vacuum-annealed TiO<sub>2</sub>(110) crystals.<sup>6,26</sup> Onishi and Iwasawa observed the formation of Ti<sub>2</sub>O<sub>3</sub> added rows on originally flat (1 × 1) terraces during O<sub>2</sub> exposure of the crystal at 800 K,<sup>6</sup> whereas Bennett *et al.* studied the reoxidation of more reduced TiO<sub>2</sub>(110) crystals, whose surfaces showed a (1 × 2) reconstruction.<sup>20,26</sup> In both of these examples, the observed changes on the surface have been explained through the reoxidation of Ti interstitials.

In another example, some of the present authors recorded STM movies at low temperatures (~187 K), showing the dissociation of water molecules in  $O_{br}$  vacancies on the TiO<sub>2</sub>(110)-(1  $\times$  1) surface.<sup>15</sup> In the same work, it was also shown that water molecules mediate the splitting of OH<sub>br</sub> pairs that resulted from water dissociation in the O<sub>br</sub> vacancies into isolated OH<sub>br</sub> (in the following, we will refer to the OH<sub>br</sub> groups as H adatoms). Regarding the dissociation of water in O<sub>br</sub> vacancies, Zhang *et al.* obtained similar results at room temperature (RT) and additionally reported that the resulting two H adatoms in the O<sub>br</sub> rows behave inequivalently.<sup>27</sup> Moreover, Zhang *et al.* revealed by means of STM that also methanol molecules dissociate in O<sub>br</sub> vacancies<sup>28</sup> and further that O<sub>br</sub> vacancies assist the diffusion of alkoxy species along the O<sub>br</sub> rows.<sup>29</sup>

Much understanding of the chemistry taking place on rutile TiO<sub>2</sub>(110) has emerged from these previous studies, highlighting the strength of STM imaging, in particular, if STM movies were successfully recorded. One may, however, object that in all these surface science studies only single reaction steps have been addressed, whereas in "real" catalytic reactions many reaction steps are involved, such as adsorption, diffusion, dissociation, recombination, and desorption. Furthermore, regarding such surface reactions wherein Obr vacancies are involved, it can be questioned whether these studies are relevant for "real" catalysis with high pressures of water and other gases. Alone, the fact that it is rather difficult to produce TiO<sub>2</sub>(110) surfaces with clean Obr vacancies [r-TiO2(110)] even under ultrahigh vacuum (UHV) conditions<sup>30–32</sup> points to a view wherein surface O vacancies are not relevant for real catalystssimply because they cannot persist under the high pressure conditions. In addition, we have recently found out that the Ti3d defect state in the band gap of reduced rutile crystals remains almost unchanged on surfaces without O<sub>br</sub> vacancies and H adatoms,<sup>33</sup> indicating that the defect state arises not because of these defects as previously believed<sup>21,22,25</sup> but rather from Ti surplus in the

near-surface region.<sup>33–35</sup> These results render the study of *r*-TiO<sub>2</sub>(110) surfaces with  $O_{br}$  vacancies merely of academic interest.

In contrast, the oxidation of H adatoms by  $O_2$  on hydrated TiO<sub>2</sub>(110) surfaces [*h*-TiO<sub>2</sub>(110)] is both of fundamental as well as practical interest<sup>36</sup> since it involves a common reactant,  $O_2$ , and a TiO<sub>2</sub>(110) surface in a realistic state without  $O_{\rm br}$  vacancies. The practical interest stems primarily from applications in areas such as heterogeneous catalysis<sup>37–39</sup> and photocatalysis<sup>24,25,36</sup> and from the need for an improved understanding of the chemical processes involved in dye-sensitized TiO<sub>2</sub>-based solar cells.<sup>23,40</sup> In most of these applications, the presence of  $O_2$  molecules is mandatory.

In this article, we show by means of high-resolution STM studies in conjunction with density functional theory calculations (DFT) that it is possible to follow the intermediate steps of this complex oxidation reaction on the  $TiO_2(110)$  surface at the atomic scale.  $O_2$ 

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molecules react with H adatoms on the  $O_{\rm br}$  rows, leading to a series of intermediate H transfer reaction steps until the formation of water is complete. STM movies provide a direct view of the reaction, revealing the intermediate steps that take place on the terraces of the oxide surface.

### **RESULTS AND DISCUSSION**

Hydrated and O2-Exposed, Hydrated TiO2(110) Surfaces. Figure 2a shows an STM image obtained after hydration of a clean reduced TiO<sub>2</sub>(110) surface that was characterized by an  $O_{hr}$  vacancy density of  $\sim$ 7% ML. In this STM image and in all the following images presented in this article, the imaging mode on the TiO<sub>2</sub>(110)-(1  $\times$  1) surface is such that the geometrically protruding Obr atoms appear as the dark rows and the Ti troughs as the bright rows.<sup>32,41,42</sup> The bright spots in the dark O<sub>br</sub> rows arise from single H adatoms<sup>27,30-32</sup> that have been formed due to water dissociation in the  $O_{br}\,vacancies^{15,22,27,32,43}$ and subsequent splitting of H adatom pairs mediated by water species.<sup>15</sup> One of the isolated H adatoms in the STM image depicted in Figure 2a is indicated by a hexagon. Additionally, the STM image in Figure 2a shows a few residual pairs of neighboring H adatoms (square), which appear brighter than single H adatoms.<sup>15,30,32</sup> Because H adatoms on next-nearest Obr atoms (rectangle, full line) appear also brighter than isolated H adatoms<sup>32</sup> and since the bare Obr atoms between two Obr atoms capped by H adatoms are not resolved, a chain of H adatoms on next-nearest O<sub>br</sub> atoms leads to the appearance of elongated bright features in the Obr rows (rectangle, broken line).

The STM image depicted in Figure 2b was acquired after 4 L O<sub>2</sub> exposure [1 L (Langmuir) =  $1.33 \times 10^{-6}$  mbar  $\cdot$  s] at ~165 K to the *h*-TiO<sub>2</sub>(110) surface from which we obtained the image shown in Figure 2a. It can be seen that new species with an elongated shape in the [001] direction have formed in the Ti troughs after the O<sub>2</sub> exposure, some of which are indicated by white circles in Figure 2b. In addition, a few very bright features (arrows) have appeared in the Ti troughs that possess STM heights as typically observed for water species.<sup>15,44</sup>

In Figure 2c, we zoom in on the indicated area in Figure 2b for a more detailed analysis. By means of a superimposed grid that was aligned in the [001] direction using the H adatoms, we checked the position of the most abundant new protrusions in the Ti troughs. Because their centers (small blue squares) lie between two intersection points along the [001] direction, it can be concluded that the new species occupy two 5f-Ti sites. The latter also follows when comparing the line profiles of the newly formed species with those of O adatoms,  $O_{ot}$ , known from previous studies<sup>30,32,33,45-47</sup> (Figure 3). Because the *h*-TiO<sub>2</sub>(110) sample was exposed to O<sub>2</sub>, the new species most likely contain two oxygen atoms; however, alone from Figures 2 and 3, it remains

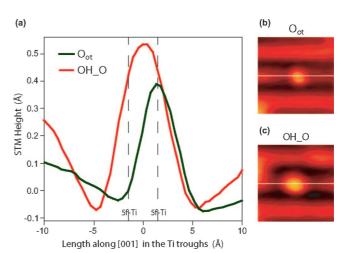


Figure 3. (a) Comparison of the STM height profiles of an  $O_{ot}$  adatom (b) and an OH\_O species (c) shown in the STM images on the right. Along the [001] direction, OH\_O species are centered between two 5f-Ti atoms unlike  $O_{ot}$  species that are centered on top of the 5f-Ti atoms. Further evidence for the assignment of the OH\_O species is given in the text.

unclear how many H atoms these new species contain. In the experiment described in Figure 2, we detected two different appearances of the new protrusions in the Ti troughs (small blue squares in Figure 2c). Both protrusions are elongated in the [001] direction but have different apparent heights. As will be discussed in the following, this result is related to the number of H atoms contained. In Figure 2c, the bigger protrusion (lower left corner) contains one H more than the species with smaller appearance (upper middle part in the image).

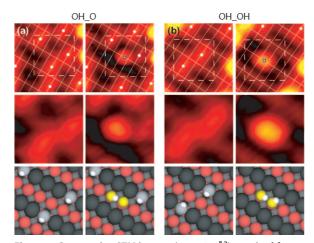


Figure 4. Consecutive STM images  $(24 \times 24 \text{ Å}^2)$  acquired from a movie (2.76 s/image) at ~190 K that show the formation of (a) OH\_O and (b) OH\_OH species. The STM movie was recorded on an *h*-TiO<sub>2</sub>(110) with a water coverage of ~6% ML and in an O<sub>2</sub> background of  $1 \times 10^{-8}$  Torr. White small dots indicate the positions of H adatoms, and small blue squares are the STM protrusions appearing due to the formation of OH\_O and OH\_OH species. Areas in the STM images indicated by white dashed squares (13 × 13 Å<sup>2</sup>) are shown enlarged in the second row and explained in the third row through ball models (top views). In the ball models, Ti atoms are shown in red and O<sub>br</sub> atoms in dark gray. The oxygen atoms of the OH\_O and OH\_OH intermediates are shown in yellow, whereas H atoms are shown as small white balls.

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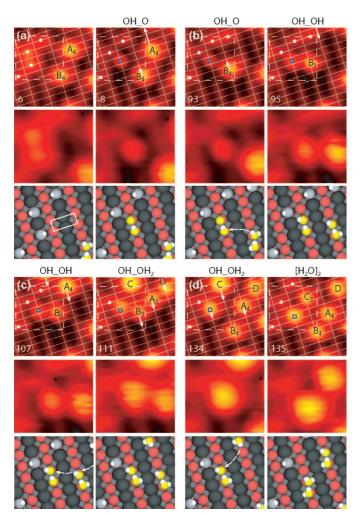


Figure 5. Snapshots extracted from an STM movie "O2\_h-TiO2\_reaction" recorded on an h-TiO<sub>2</sub>(110) surface with a water coverage of  $\sim$  6% ML and in an O<sub>2</sub> background of  $1 \times 10^{-8}$  Torr. The STM images shown in the first row are  $30 \times 30$  Å<sup>2</sup>, but the whole scanned area in the movie was  $84 \times 84 \text{ Å}^2$ . Images were acquired with a rate of 2.76 s per frame. OH\_O formation is shown in (a), and the three following H uptake reactions are identified in (b) OH\_OH formation, (c) OH\_OH<sub>2</sub> formation, and (d) [H<sub>2</sub>O]<sub>2</sub> formation, respectively. Numbers at the lower left edges indicate the appearance of the selected STM images within the movie. White small dots indicate H adatoms, and water species in the form of dimers are denoted using the labels A<sub>n</sub>, B<sub>n</sub>, C<sub>n</sub>, and D, respectively, where n indicates the estimated number of containing H atoms (*i.e.*,  $A_n$  with n = 4 denotes a stoichiometric water dimer). In (c) and (d), the index n was omitted for some water dimers. The lattice grid is centered on 5f-Ti sites. White arrows along the Ti troughs indicate the diffusion directions of the adsorbates. Indicated areas of  $18 \times 16.5 \text{ Å}^2$  size in the STM images in the first row (white dashed rectangles) are shown enlarged in the second row and explained in the third row using ball models (top views). Ti atoms are shown in red and O<sub>br</sub> atoms in dark gray. The oxygen atoms of the forming water dimer  $(O_{ad})$  are shown in yellow, whereas H atoms are shown as small white balls. The bent arrows indicate from which side in the STM images the H adatoms were provided.

### **Observations in STM Movies Acquired on Partly Water-Covered**

*h*-TiO<sub>2</sub>(110). To gain more detailed information about the assignment of these new species, we recorded an STM movie on a large terrace on the *h*-TiO<sub>2</sub>(110) surface. The STM movie was started before the inlet of O<sub>2</sub> gas through a leak valve was initiated. When stable scanning conditions were achieved, the O<sub>2</sub> background pressure in the chamber was increased and stabilized

at  $\sim 10^{-8}$  Torr. In the STM movie, we observed several events as those depicted in Figure 4, where new protrusions appeared in the Ti troughs concomitant with the disappearance of H adatoms on adjacent O<sub>br</sub> rows. In the first event (Figure 4a), only one H adatom disappeared, whereas in the second event (Figure 4b), two H adatoms disappeared simultaneously as one protrusion appeared in the Ti trough. Events where only one H adatom disappears were observed 10 times during the course of the STM movie, whereas the event shown in Figure 4b where two H adatoms disappear was observed only once. On the basis of these results, we assign the elongated species seen in Figure 2b,c to intermediates with HO<sub>2</sub> and H<sub>2</sub>O<sub>2</sub> stoichiometry. This conclusion is consistent with an analysis of the full data set corresponding to the images shown in Figure 2a,b since the total amount of hydrogen before and after the O<sub>2</sub> exposure was found to be identical if we assign the newly found species as species with HO<sub>2</sub> and H<sub>2</sub>O<sub>2</sub> stoichiometry (Figure 2d). In the following, we denote the intermediates as OH\_O and OH\_OH, respectively, because, according to the DFT calculations presented below, the O-O bonds are broken in both of these intermediate species.

We may add that the appearance of the OH\_O and OH\_OH species in the STM images leads to changes of the STM heights of other adsorbates and of the O<sub>br</sub> rows. For example, the STM height of the remaining H adatom (right panel in Figure 4a) in the vicinity of the OH\_O species is clearly lower than when the OH\_O species is not nearby (left panel in Figure 4a). Likewise, the STM height of the Obr rows drops in the vicinity of OH\_O and OH\_OH species, leading to a sombrero-like shape of the adsorbates in the STM images (right panels in Figure 4a,b, respectively). A sombrero-like shape is also evident for the O<sub>ot</sub> species (Figure 3), suggesting that the decreased STM height in the vicinity of O-containing adsorbates is related to the electronegative character of the adsorbates. These results further underline the now well-established fact that the STM images obtained on the  $TiO_2(110)$  surface are strongly influenced by electronic effects.<sup>32,41,42,44</sup>

In the spanned temperature range from ~180 to ~200 K, we observed in the STM movies that the adsorbed OH\_O and OH\_OH species were immobile. Likewise, the H adatoms that could further react with these intermediates were immobile in this temperature range. This observation suggests that kinetics explains the apparent stability of the OH\_O and OH\_OH species under the given conditions. However, we know from previous studies that water mediates the net diffusion of H adatoms on *h*-TiO<sub>2</sub>(110).<sup>15</sup> In ref 15, the diffusion of H adatoms was reported to occur perpendicularly to the O<sub>br</sub> rows. Here we observe that under certain conditions water species also mediate the diffusion of H adatoms along the O<sub>br</sub> rows. In any case, a trace amount of

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adsorbed water may facilitate further reactions on the h-TiO<sub>2</sub>(110) surface.

In another part of the STM movie (the STM movie "O2\_h-TiO2\_reaction" can be accessed at www.phys.au.dk/spm/movies/O2\_h-TiO2\_reaction.gif and the Supporting Information), the newly formed OH\_O and OH\_OH species reacted further in the presence of coadsorbed water species. From an analysis of the STM images acquired before the O<sub>2</sub> exposure,  $\sim 6\%$ of the 5f-Ti sites were occupied by H<sub>2</sub>O, which was dosed directly after hydration of the  $TiO_2(110)$  surface. In Figure 5, water species in the form of dimers are denoted using the labels  $A_n$ ,  $B_n$ ,  $C_n$ , and D, respectively, where n indicates the estimated number of bound H atoms (*i.e.*,  $A_n$  with n = 4 denotes a stoichiometric water dimer). Starting from OH\_O formation, Figure 5a, we identified three additional H uptake reactions: (i) OH\_OH formation (Figure 5b), (ii) OH\_OH<sub>2</sub> formation (Figure 5c), and finally, (iii) the creation of a water dimer,  $[H_2O]_2$  (Figure 5d). Interestingly, we found in the STM movie that upon OH\_OH formation the second H stems from a water dimer (denoted as  $B_4/B_3$  in Figure 5b).

It is possible from the high-resolution STM images to identify these reaction intermediates because, upon H uptake (*i.e.*, when OH\_O changes to OH\_OH and when OH\_OH changes to OH\_OH<sub>2</sub>), the protrusions seen in the STM images always increased (Figure 6). Furthermore, the diffusion rate of the intermediates along the Ti trough differs markedly depending on how many H atoms the intermediate contains. The latter we revealed from several STM movies, one of which was used to extract the data shown in Figure 7. In the experiment corresponding to the data shown in Figure 7, we first exposed an h-TiO<sub>2</sub>(110) surface to O<sub>2</sub> at RT to decrease the number of H adatoms on the surface<sup>33</sup> before a small amount of water was dosed at  $\sim$  173 K. In the presented typical example, the mobility of the OH OH<sub>2</sub> and water dimer species differs by about a factor of 30, which allows us also to differentiate between these two species in the STM movies in spite of their similar STM heights.

We can conclude that the reaction product observed in movie "O2\_h-TiO2\_reaction" is indeed water in the form of water dimers by comparing the number of reacted O<sub>2</sub> molecules with the number of reacted H adatoms. Figure 8a depicts an STM image from the initial STM movie recorded before the O<sub>2</sub> exposure was initiated. In this STM image, we find a number of water dimers and  $\sim$ 46 H adatoms. The STM image depicted in Figure 8b shows the same surface area as in Figure 8a but 1297 images, that is, 1 h later. While scanning and running the reaction for 1 h, the formation of seven intermediate OH\_O species and their further reaction were observed, leading to an increased number of water dimers. Water dimers also diffused into the scanned area from the surrounding area, leading to additional increase in the number of water dimers. Simulta-

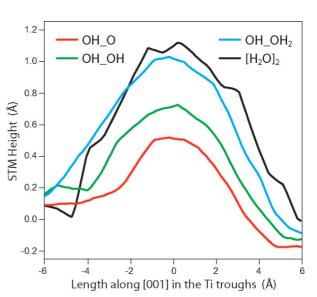


Figure 6. STM height profiles of the OH\_O, OH\_OH, OH\_OH<sub>2</sub>, and  $[H_2O]_2$  species corresponding to the images shown in Figure 5a, right panel, Figure 5b, right panel, Figure 5d, left panel, and Figure 5d, right panel.

neously, the number of H adatoms in the same surface area has decreased to  $\sim\!15$  (Figure 8b). This result implies that  $\sim\!4.4$  H adatoms have disappeared per  $O_2$  molecule, which is very close to the expected H-to- $O_2$  ratio of 4:1 according to the proposed reaction pathway.

Further STM Experiments Addressing 0<sub>2</sub>-Exposed Ti0<sub>2</sub>(110) Surfaces. Because water species in the submonolayer range are known to desorb from vacuum-annealed TiO<sub>2</sub>(110) surfaces at ~RT,<sup>36,46,48</sup> newly formed water species, as for instance observed in the STM movie discussed above, should be absent after heating of the sample to temperatures higher than RT. In fact, this situation can be achieved, as is seen from the results depicted in Figure 9. The STM image shown in Figure 9a was acquired subsequently to the experiment de-

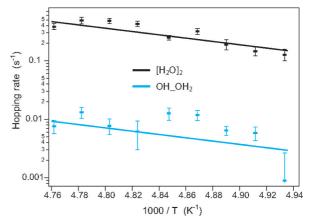


Figure 7. Arrhenius plots for  $OH_OH_2$  and  $[H_2O]_2$  species on almost perfect areas on the  $TiO_2(110)$  surface. To extract the hopping rates for diffusion along the Ti troughs as a function of temperature only such diffusion events of the  $OH_OH_2$  and  $[H_2O]_2$  species were considered when no other species were nearby. The identification of the species rests on typical H transfer and uptake reactions as obtained for the two species considered.

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T ~192 K

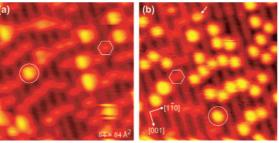


Figure 8. Two STM images acquired within the original STM movie from which we extracted the data presented in Figures 4 and 5 and movie "O2\_h-TiO2\_reaction". The STM image in (a) was acquired before the O<sub>2</sub> exposure was initiated and at  $\sim$ 186 K. At this time, the number of H adatoms in the scanned area was 46. (b) STM image showing the same surface area as in (a) but 1297 images, that is, 1 h later than the image depicted in (a). This image was acquired with the sample at  $\sim$ 192 K. Symbols indicate H adatoms on O<sub>br</sub> sites (hexagon) and water dimers in the Ti troughs (circle), respectively. The arrow in (b) indicates a water monomer.

scribed in Figure 2a,b and after heating of the crystal up to  $\sim$ 376 K. Compared to the initial density of H adatoms (~14.2% ML), the density of H adatoms has strongly decreased to  $\sim$ 4.2% ML after heating of the crystal (cf. Figure 9b). It appears that the OH\_O formation was not complete at 165 K since the density of H adatoms measured directly after the O<sub>2</sub> exposure at 165 K was only slightly decreased. Furthermore, we emphasize that in this experiment, where the O<sub>2</sub> exposure was rather low, no other species in addition to the residual H adatoms were left on the surface after heating. Specifically, we did not find any O<sub>ot</sub> adatoms or OH\_O species on the TiO<sub>2</sub>(110) surface (Figure 9a). This result indicates that in the experiment considered all the oxygen atoms that stem from the initially adsorbed and reacted O<sub>2</sub> molecules have left the surface upon heating, most likely in the form of water dimers.

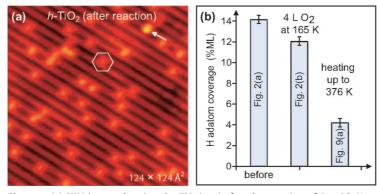


Figure 9. (a) STM image showing the TiO<sub>2</sub>(110) after the reaction of O<sub>2</sub> with H adatoms. To initiate the reaction, the *h*-TiO<sub>2</sub>(110) surface as depicted in Figure 2a was exposed to 4 L O<sub>2</sub> at ~165 K and then slowly heated up to ~376 K (ramp ~2 K/s). Since the O<sub>2</sub>-exposed *h*-TiO<sub>2</sub>(110) was inspected by STM (see Figure 2b), it is very likely that trace amounts of water from the background have adsorbed on the surface before the temperature was ramped up to ~376 K. The STM image was acquired at ~110 K. Symbols indicate isolated H adatoms (hexagon) and a water monomer (arrow). (b) Bar graph of measured coverages in %ML of H adatoms before (*cf.* Figure 2a), after O<sub>2</sub> exposure (*cf.* Figure 2b), and after the subsequent heating up to ~376 K. To determine the densities of H adatoms, areas of ~2000 nm<sup>2</sup> were scanned and analyzed.

Finally, we studied whether Oot adatoms can be reacted off from the  $TiO_2(110)$  surfaces when interacting with coadsorbed water and H adatoms. To ensure the availability of H adatoms on the surface, we started with a partly oxidized TiO<sub>2</sub>(110) surface that was characterized by the presence of Oot adatoms and Obr vacancies (Figure 10a). To prepare such a partly oxidized surface, a clean r-TiO<sub>2</sub>(110) surface was exposed to 8 L O<sub>2</sub> at RT by backfilling the UHV chamber. In addition to O<sub>ot</sub> adatoms and O<sub>br</sub> vacancies, we also find H adatoms on the TiO<sub>2</sub>(110) surface. The latter comes about because of the interaction of O<sub>2</sub> molecules with the walls (and other parts) of the UHV chamber, which inevitably leads to an increased partial pressure of water. After water exposure corresponding to a coverage of  $\sim 10\%$  ML at  $\sim$ 113 K followed by slowly heating of the crystal up to  $\sim$  343 K, the result was an *h*-TiO<sub>2</sub>(110) surface with a low density of H adatoms (Figure 10b). From this experiment, it is very clear that-through the formation of water-indeed O<sub>ot</sub> adatoms can be reacted off from the TiO<sub>2</sub>(110) surface.

**DFT Calculations.** To verify the reaction pathway revealed from the STM movie and to clarify the nature of the reaction intermediates, we performed first-principles DFT calculations using the DACAPO package,<sup>49</sup> ultrasoft pseudopotentials, and the revised Perdew–Burke–Ernzerhof (RPBE) exchange– correlation functional.<sup>50</sup> The slabs used in the calculations were composed of six TiO<sub>2</sub> trilayers with  $c(4 \times 2)$  surface cells and  $2 \times 2 k$ -points. In order to model bulk-reduced samples, a crystallographic shear-plane (CSP)<sup>51,52</sup> parallel to the (110) surface was introduced by removing every second O atom between the fourth and fifth layer of Ti atoms and by shifting the upper four trilayers in the (1)/(2)[01] direction. The distance of the CSP from the (110) surface was chosen such that

the energetics of the  $O_2$  adsorption on the surface compares to that of  $O_2$  adsorption on a six trilayer slab containing an isolated Ti interstitial.<sup>33</sup>

Figure 11 presents the calculated energetics (Figure 11a) and adsorption configurations of all reaction intermediates (Figure 11b-h). Starting from the *r*-TiO<sub>2</sub>(110) surface (E = 0) with two O<sub>br</sub> vacancies and two H<sub>2</sub>O molecules in the gas phase, energy is released upon water dissociation in O<sub>br</sub> vacancies, leading to two H adatoms per O<sub>br</sub> vacancy. The potential energy curve continues to go "down hill" upon subsequent O<sub>2</sub> adsorption (Figure 11b), OH\_O and OH\_OH formation, and further all the way to the final reaction product-a water dimer, [H<sub>2</sub>O]<sub>2</sub> (Figure 11h). These DFT calculations are consistent with the presented STM results, as well as with previously published spectroscopic evidence of water formation by Henderson et al.<sup>36</sup> Both the experimental results as well as the computed energetics presented here are at variance with published theoretical work addressing O2 adsorption

on the h-TiO<sub>2</sub>(110) surface.<sup>37,53</sup> This discrepancy we trace back primarily to the fact that the authors of refs 37 and 53 assumed a stoichiometric TiO<sub>2</sub> bulk without extra electrons available.

We also calculated the activation energy barriers,  $E_{ar}$  of interest for the proposed reaction sequence. The transfer of the first H adatom from a position on an  $O_{br}$  atom to  $O_2$  (Figure 11b) is hindered by a moderate barrier of  $E_a = 0.34$  eV. The resulting  $\pi$ -bonded HO<sub>2</sub> species (Figure 11c) instantly ( $E_a = 0.05$  eV) transforms into a  $\sigma$ -bonded HO<sub>2</sub> species (Figure 11d) that dissociates very easily ( $E_a = 0.15$  eV). According to the low barriers for the latter two reaction steps, any HO<sub>2</sub> species is very shortlived even at the low temperatures used for the STM measurements. Therefore, our DFT calculations suggest that all reaction intermediates observed in the STM images show species in which the O–O bond has been broken.

In Figure 12, we present simulated STM images using the Tersoff–Hamann formalism<sup>54</sup> of four selected configurations, which we considered in Figure 11. It can be seen that the apparent heights in the simulated STM images mirror very well the ones obtained experimentally. Likewise, the elongated shape in the [001] direction is well reproduced in the STM simulations. It should be noted, however, that in the experiments an average over many different configurations of similar or slightly higher energies is seen, whereas the simulations show the local density of states (DOS) only for one particular configuration. This is one of the reasons why all the simulated STM images show an asymmetry, including those corresponding to the OH\_OH species (cf. Figure 12b) and to the water dimer (cf. Figure 12d), whereas the experimental STM images are symmetric about the [001] direction. Of course, for the configurations shown in Figure 12a,c with a H adatom (H<sub>ad</sub>) on an Obr atom nearby, symmetry about the [001] direction cannot be expected. Within the time scale of STM imaging (several seconds per image), the latter two configurations, OH\_O plus a H<sub>ad</sub> (cf. Figure 12a) and H<sub>2</sub>O\_OH plus a H<sub>ad</sub> (cf. Figure 12c, respectively, cannot be observed. Nevertheless, the presented set of simulated STM images for a series of adsorbates shows that we are now approaching a situation that allows a direct comparison between experimental and theoretical results also for semiconducting TiO<sub>2</sub>.

**Comparison with Temperature-Programmed Desorption Measurements.** The presented STM and DFT results are in good agreement with previously published data using temperature-programmed desorption and electron energy loss measurements.<sup>36</sup> Chemisorbed OH\_O is the initial intermediate that is produced *via* H transfer reactions from positions on the O<sub>br</sub> rows to O<sub>2</sub> molecules, which concomitantly dissociate. With subsequent H transfers, OH\_OH species are formed in the Ti troughs, in agreement with previous suggestions<sup>36</sup> [note that in

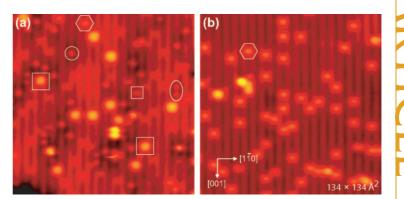


Figure 10. (a) STM image showing a partly oxidized TiO<sub>2</sub>(110) surface that was prepared by an O<sub>2</sub> exposure of 8 L at RT to a clean, reduced TiO<sub>2</sub>(110) surface. (b) STM image obtained after water exposure corresponding to a coverage of ~10% ML at ~113 K, followed by slowly heating of the crystal up to ~343 K. Symbols indicate O<sub>br</sub> vacancies (small square), isolated H adatoms (hexagon), paired H adatoms (big square), isolated O<sub>ot</sub> adatoms (circle), and pairs of next-nearest O<sub>ot</sub> adatoms (ellipse).

refs 36 and 46 the OH\_OH species are denoted as pairs of terminal OH groups (OH<sub>t</sub>)]. However, as shown in the present work, the further oxidation can proceed differently than proposed by Henderson *et al.*<sup>36</sup> Mediated by coadsorbed water species that make the diffusion of

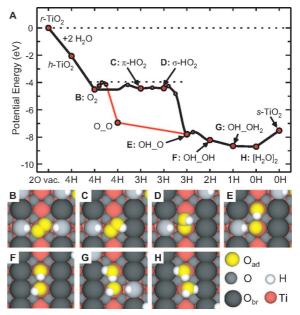


Figure 11. (a) DFT-based potential energy diagram. Big red dots correspond to calculated local potential energy minima, whereas small red dots indicate values deduced using the climbing NEB procedure. The initial configuration is a reduced  $TiO_2(110)$  surface [r-TiO\_2(110)] with two O<sub>br</sub> vacancies and two H<sub>2</sub>O molecules in the gas phase. The first step is dissociation of the H<sub>2</sub>O molecules in O<sub>br</sub> vacancies, thereby forming four H adatoms [h-TiO<sub>2</sub>(110)] (the number of O<sub>br</sub> vacancies or H adatoms on the  $TiO_2(110)$  surface is denoted on the bottom axis). The second step is the adsorption of an O<sub>2</sub> molecule on h-TiO<sub>2</sub>(110). The pathway in which O<sub>2</sub> becomes HO<sub>2</sub> before dissociation into OH O is shown as a thick black line, whereas the thin red line accounts for a pathway in which O<sub>2</sub> dissociates ( $E_a = 0.51$  eV) before the reduction. The former pathway is kinetically preferred. Desorption of the [H<sub>2</sub>O]<sub>2</sub> leads to a clean TiO<sub>2</sub>(110) surface, s-TiO<sub>2</sub>(110), with no point defects in the uppermost surface layer. (b-h) Ball-and-stick models (top views) of selected reaction intermediates. O<sub>ad</sub> (yellow balls) indicates the O atoms of the adsorbates.

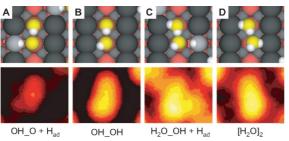


Figure 12. Simulated STM images for selected intermediates together with the corresponding ball-and-stick models: (a) OH\_O plus a H adatom (H<sub>ad</sub>) nearby (cf. Figure 11e), (b) OH\_OH (cf. Figure 11f), (c) H<sub>2</sub>O\_OH plus a H<sub>ad</sub> nearby (cf. Figure 11g), and (d) [H<sub>2</sub>O]<sub>2</sub> (cf. Figure 11h). In the simulation, the local density of states (DOS) was constructed from electronic states in an energy window between 0.25 and 0.75 eV above the Fermi level (empty states). Isosurfaces of the local DOS are shown for  $3 \times 10^{-9}$  states per Å<sup>3</sup>. The color map covers heights in a 2 Å interval; black: ~4 Å, red: ~4.8 Å, orange: ~5.2 Å, yellow: ~5.6 Å, and white: ~6 Å.

H adatoms to O<sub>br</sub> sites next to the OH\_OH species possible, the reaction continues *via* H transfers to the OH\_OH species, which finally leads to the formation of water dimers.

Previously, Henderson et al. proposed that the OH\_OH species decompose into water monomers and O<sub>ot</sub> adatoms, and that the water monomers desorb at RT, whereas the O<sub>ot</sub> adatoms remain on the surface.<sup>46</sup> If this were the only possible scenario, heating of an O<sub>2</sub>exposed h-TiO<sub>2</sub>(110) surface to temperatures higher than RT should always lead to a TiO<sub>2</sub>(110) surface with O<sub>ot</sub> adatoms, irrespectively of the amount of O<sub>2</sub> molecules dosed. As evidenced by the STM image depicted in Figure 9a, no Oot adatoms are left on the surface if small O<sub>2</sub> exposures are used. This means that, as long as sufficient H adatoms are available, the water formation will be completed and thus no Oot adatoms or OH\_O species are left on the surface after heating of the crystal to temperatures higher than RT. Despite the fact that the oxidation reaction of H adatoms by O<sub>2</sub> is self-accelerating (the newly formed water species also facilitate the transport of H adatoms toward O<sub>2</sub> and the intermediate species), trace amounts of water must be available on the surface to complete the water formation reaction since without water the H adatoms are immobile at temperatures as high as 200 K.

The presented results are remarkable for at least two reasons. First, the reaction mechanism revealed here for the oxidation reaction of H adatoms by O<sub>2</sub> on TiO<sub>2</sub> mediated by coadsorbed water is interesting, particularly with a view on the promoting role of moisture for catalytic reactions on supported gold clusters<sup>38,55</sup> and the strong indications that the activation of molecular oxygen is the most important reaction step on supported Au nanocatalysts.<sup>37,38</sup> Second, the observed reaction mechanism has presumably universal character for TiO<sub>2</sub> surfaces, meaning that the reaction probably proceeds similarly on other faces of rutile and maybe also on other reducible transition metal oxides. Comparing the available spectroscopic data for  $TiO_2(110)$ with those acquired on the (110) face of RuO<sub>2</sub><sup>56,57</sup>—an oxide surface that is isomorphous to the  $TiO_2(110)$ surface-some striking similarities are revealed. On both of these surfaces, the transfer of H adatoms from O<sub>br</sub> sites to oxygen species in the troughs is facile, and in both cases, the formation of water has been observed.

### CONCLUSIONS

Unprecedented details about the chemical reaction of  $O_2$  with H adatoms on rutile TiO<sub>2</sub>(110) have been revealed by an interplay of time-lapsed, high-resolution STM studies and DFT calculations.  $O_2$  molecules react with H adatoms on the  $O_{br}$  rows, leading to a series of intermediate H transfer reaction steps until the formation of water is complete. In order to make the completion of the water formation reaction at temperatures below 200 K possible, the coadsorption of trace amounts of water is required to generate low-barrier pathways for the diffusion of H adatoms. We have shown how the STM with its unique spatial resolution and versatility literally allows us to "watch" a chemical reaction on an oxide surface of high interest.

### **METHODS**

**Experimental Details.** The experiments were carried out in an ultrahigh vacuum (UHV) chamber with a base pressure in the low  $10^{-11}$  Torr range equipped with a home-built, temperature-variable STM<sup>58,59</sup> and standard facilities for sample preparation and analysis.<sup>60</sup> The sample temperature could be varied from 100 K (by cooling with liquid N<sub>2</sub>) to 1200 K by radiative heating and electron bombardment from the back side of the sample; the temperature was measured by a K-type thermocouple spot welded on the sample plate close to the crystal. If the samples were warmed to certain temperatures, we throughout kept the sample for 2 min at the intended temperature to ensure that the crystals reached the temperature measured at the sample plate.

The TiO<sub>2</sub>(110)-(1  $\times$  1) samples used for these STM studies were cleaned by several cycles of Ar<sup>+</sup> bombardment at RT and vacuum annealing at 800–950 K. The density of bridging oxy-

gen (O<sub>br</sub>) vacancies was in the range of 6–8% ML, with 1 ML (monolayer) being the density of the (1 × 1) unit cells, 5.2 × 10<sup>14</sup>/ cm<sup>2</sup>. TiO<sub>2</sub>(110) surfaces with H adatoms [*h*-TiO<sub>2</sub>(110)] were prepared by water exposure at temperatures ≤170 K to a reduced crystal, followed by slowly warming the crystal up to 373 K. Water exposures were performed *via* background exposure through a leak valve. Several freeze–pump–thaw cycles were performed to purify the deionized water. If not stated otherwise, O<sub>2</sub> was dosed *via* backfilling of the chamber, using a partial O<sub>2</sub> pressure of maximal 5.0 × 10<sup>-8</sup> Torr. A cooling trap at the O<sub>2</sub> gas line was used to avoid moisture during O<sub>2</sub> exposure.

The STM images presented in this work were acquired in the constant current mode using a tunneling voltage ( $V_T$ ) of +1.25 V and a tunneling current ( $I_T$ ) of  $\sim$ 0.1 nA. A special preamplifier was used, allowing us to work with this low tunnel current and with high feedback gains. Throughout electrochemically etched tungsten tips were used for the STM measurements.

SNANO

To extract the hopping rates from the STM movies, we first determined so-called displacement distributions from the movies for all the species considered. The displacement distributions, that is, the probabilities for the species to jump 0, 1, 2 *etc.* sites between two successive STM images, were found by comparing the positions of the different species in successive STM images in the STM movies. From these displacement distributions, the hopping rates of the species can be deduced, as described by Ehrlich.<sup>61</sup> For the analysis of the [H<sub>2</sub>O]<sub>2</sub> diffusion, an average of 540 events was used for each point in the Arrhenius plot, for example, 10 different [H<sub>2</sub>O]<sub>2</sub> species which were followed on 55 consecutive STM images on average. For the analysis of the OH\_OH<sub>2</sub> diffusion, on average, 560 events were used for each of the points in the Arrhenius plot.

**Computational Details.** The first-principles density functional theory (DFT) calculations were performed using the DACAPO package<sup>50,62</sup> employing plane waves and ultrasoft pseudopotentials. The Ti3*p* states were treated as core states since tests with valence Ti3*p* states showed the energetics to change only little. Exchange – correlation effects were described by the RPBE functional.<sup>50</sup> Slabs of six TiO<sub>2</sub> trilayers with *c*(4 × 2) surface cells and 2 × 2 *k*-points were used. The one-electron states were populated according to a Gaussian broadening scheme with *k*<sub>B</sub>*T* = 0.1 eV. All atoms in the super cell were fully relaxed, and the climbing nudged elastic band (NEB) method<sup>63</sup> was used for determining transition states.

In order to model bulk-reduced samples, a crystallographic shear-plane (CSP)<sup>51,52</sup> parallel to the (110) surface was introduced by removing every second O atom between the fourth and fifth layer of Ti atoms and by shifting the upper four trilayers in the (1)/(2)[011] direction. The distance of the CSP from the (110) surface was chosen such that the energetics of the O<sub>2</sub> adsorption on the surface compares to that of O<sub>2</sub> adsorption on a six trilayer slab with isolated Ti interstitials.<sup>33</sup>

In the calculations addressing adsorbed OH and HO<sub>2</sub> species, we tested the possible influence of spin polarization. However, since the substrate has electrons in gap states right below the conduction band edge, the empty spin states of the adsorbed OH and HO<sub>2</sub> species become filled, leading to vanishing of their net spin moments. Therefore, in subsequent calculations, spin polarization effects could be ignored.

The simulated STM images were constructed as topographs of constant local density of states (3  $\times$  10<sup>-9</sup> states per Å<sup>3</sup>) according to the Tersoff–Hamann procedure.<sup>54</sup> The local density of states was constructed from electronic states in an energy window between 0.25 and 0.75 eV above the Fermi level (*i.e.*, empty states). The color map covers heights in a 2 Å interval; black: ~4 Å, red: ~4.8 Å, orange: ~5.2 Å, yellow: ~5.6 Å, and white: ~6 Å.

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Supporting Information Available: STM movie. This material is available free of charge via the Internet at http://pubs.acs.org.

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